

Synthesis of minerals

Iskrina A.V.¹, Spivak A.V.¹, Setkova T.V.¹, Khasanov S.S.², Kuzmin A.V.², Zakharchenko E.S.¹, Viryus A.A.¹ Synthesis and structural characteristics of a new compound GaGeO₃OH - an analogue of the Egg phase AlSiO₃OH.

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Abstract. We present a complex study (scanning electron microscopy, energy dispersive X-ray spectroscopy, powder X-ray diffraction, and Raman spectroscopy) on the synthetic GaGeO₃OH compound. For the first time, the GaGeO₃OH crystals (up to 10 μm in size) are synthesized at 7 GPa and 1000 °C. The unit cell parameters are: $a = 7.5785(1) \text{ \AA}$, $b = 4.4605(3) \text{ \AA}$, $c = 7.2469(4) \text{ \AA}$, $\beta = 97.519(2)^\circ$, $V = 242.87(02) \text{ \AA}^3$, space group is $P2_1/n$, and are characterized by large values compared to the natural and synthetic Egg phase. The experimentally obtained Raman spectrum of GaGeO₃OH has a topology similar to the Raman spectra of the Egg phase (natural and synthetic AlSiO₃OH and model MgSiO₄H₂). In the high-frequency region of the Raman spectra, a group of wide OH⁻ vibration bands is recorded, which indicates the presence of several hydrogen positions in the crystal structure of GaGeO₃OH.

Keywords: phase Egg, synthesis, gallium, germanium, crystal structure

Introduction. One of the relevant areas of experimental mineralogy, petrology, and geochemistry is the study of high-pressure phases of possible H₂O reservoirs in the mantle transition zone. Within the framework of the Al₂O₃ – SiO₂ – H₂O model system, most of the research has focused on studying the Egg phase, first described by Eggleton et al. (1978). Later, this phase was discovered in inclusions in ultra-deep diamonds (Wirth et al., 2007). The crystal structure of AlSiO₃OH, including the position of hydrogen, was solved and refined using high-resolution X-ray powder diffraction: $a = 7.14409(2) \text{ \AA}$, $b = 4.33462(1) \text{ \AA}$, $c = 6.95253(2) \text{ \AA}$ and $\beta = 98.396(1)^\circ$, $Z = 4$, $V_0 = 212.99(1) \text{ \AA}^3$, space group is $P2_1/n$ (Schmidt et al., 1998). The model end member of the MgSiO₄H₂ solid solution ($P2_1/n$) with AlSiO₃OH, according to the substitution of $\text{Al}^{3+} \leftrightarrow \text{Mg}^{2+} + \text{H}^+$, was synthesized at 24 GPa and 1400 °C (Bindi et al., 2020). This paper presents the results of the synthesis of a new compound GaGeO₃OH, an analog of the Egg phase AlSiO₃OH, as well as data on the refinement of its crystal structure.

Methods. The synthesis experiments were carried out using a toroidal anvil apparatus at the D.S. Korzhinsky Institute of Experimental Mineralogy of the Russian Academy of Sciences

(IEM RAS). The toroidal anvil apparatus is a modification of the Bridgman anvil apparatus (Litvin, 1991). A powdered mixture of gallium (99.5%) and germanium (99.9%) oxides was used as the starting material. The oxides were mixed in a 1:1 ratio and placed in a Pt ampoule with the addition of excess deionized water. The experiment was carried out at $P = 7.0 \text{ GPa}$ and $T = 1000^\circ\text{C}$ for 6 hours. The pressure was determined using the standard Bi polymorphisms (at 2.55, 2.7, and 7.7 GPa), Tl (at 3.67 GPa), and Ba (5.5 GPa) with an accuracy of 0.25 GPa. The temperature was controlled and measured by a thermocouple (Pt₇₀Rh₃₀/Pt₉₄Rh₆) with an accuracy of $\pm 20^\circ\text{C}$. After the experiment, the sample was cooled to 20°C at a rate of $300 \pm 50^\circ\text{C/s}$.

The morphology of the obtained crystals and composition were studied using a Tescan Vega II XMU scanning electron microscope (SEM) using secondary and reflected electron detectors, as well as an energy-dispersive X-ray spectrometer (INCAx-sight) for quantitative analysis. Electron microprobe analysis (EMPA) of the studied samples was performed at an accelerating voltage of 20 kV, an absorbed electron current of 0.2 nA, and a probe size of 0.3 μm. The analysis time was 70 s. The following standards were used for the quantitative determination of the main elements: for Ge – elemental germanium, for Ga – GaP.

The Raman spectra of the crystals were measured on a setup consisting of an Acton SpectraPro-2500i spectrograph with a Pixis2K CCD detector cooled to -70°C and an Olympus microscope with a continuous solid-state monomeric laser with a wavelength of 532 nm. The laser beam was focused on the sample using an Olympus 50× lens into a spot with a diameter of $\sim 5 \text{ \AA}$. Raman data were collected by repeated exposures of 540 s ($3 \times 180 \text{ s}$). The software Fityk 1.3.1 was used for profile fitting for the Raman band analysis.

Powder X-ray diffraction data were collected with BRUKER D8Advance using capillary scheme at ambient conditions (atmospheric pressure, 25°C). The 0.30 mm capillary was loosely filled with a fine powder of the synthesized phase. Johanson monochromator with CuKα₁ radiation and with focus on the detector was used for the primary beam. In addition to sharp diffraction peaks, the spectrum contained a broad diffuse maximum resulting from scattering at the amorphous capillary material. To better approximate the background of the spectrum, this diffuse maximum was subsequently considered as a contribution from the amorphous minor phase.

The resulting peak profiles and the refinement of the peak positions and all spectra, the determination

of the lattice parameters and the related indexing of the observed diffraction maxima were carried out according to the standard procedures of the TOPAS software Ver. 4.2 (Bruker AXS, Topas V4, 2008). Special attention was paid to the accuracy of the refinement of the spectral background, since many weak reflections were observed. The obtained values of the lattice parameters and the characteristics of the diffraction spectrum were used in the solution of the structure. It should be noted that the indexing and unit cell determination procedure was stable with different choices of the first 25–40 reflexes, with different levels of clipping of weak peaks, and with different variants of the analytical form of maxima.

To solve the structure, the method of simulated annealing was used in the mode of comparison of experimental and theoretical spectra using the advanced features of the TOPAS software package. The composition of the cell was set as 4 Ga atoms, 4 Ge atoms and 16 oxygen atoms, according to the data of elemental analysis and space filling analysis. In accordance with the symmetry group $P2_1/n$ (below), the coordinates of the independent part of the structure, Ga1Ge1O_4 , were determined. The structure solution routine quickly comes to a stable configuration of atoms (Ga and Ge atoms in an octahedral coordination of oxygen atoms) with a confidence factor of about $R_{\text{wp}} = 17\%$.

The resulting structure model was further refined without constraints by using the standard method of full-profile fitting of X-ray diffraction spectra (Rietveld method) in the TOPAS software to the reliability factors $R_{\text{wp}} = 4.05\%$, $R_p = 3.06\%$ and $R_{\text{exp}} = 2.86\%$, consequently $\chi^2 = 1.42$, and $R_{\text{Bragg}} = 1.47\%$, which is a good indicator of the confidence of the final structure model.

Results and discussion. As a result of the experiments, a fine crystalline powder is formed in the entire volume of the capsule. White crystals have the isometric shape of a pseudocubic habitus. The size of individual crystals reaches 5–8 microns (Fig. 1). The chemical composition is defined as Ga_2O_3

40.9 ± 0.8 , GeO_2 $43.8 \pm 0.8\%$ by weight. The crystals used for X-ray diffraction are homogeneous within the limits of analytical error.

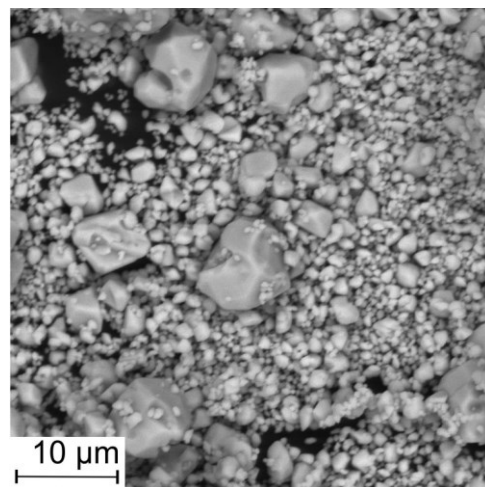


Fig. 1. SEM images of synthesized GaGeO_3OH crystals.

X-ray diffraction analysis revealed that the GaGeO_3OH phase crystallizes in the $P2_1/n$ (14) space group with unit cell parameters: $a = 7.5785(1) \text{ \AA}$; $b = 4.4605(3) \text{ \AA}$; $c = 7.2469(4) \text{ \AA}$; $V = 242.87(02) \text{ \AA}^3$; $\beta = 97.519(2)$. The structure of GaGeO_3OH consists of GeO_6 octahedra connected by common edges and a Ga_2O_{10} dimer, similar to the pure Egg phase of AlSiO_3OH (Schmidt et al., 1998) (Fig. 2). The connection of these elements to form the structure of the Egg phase is similar to the connection of elements of another high-pressure phase, stishovite, in which octahedra connected by edges form columns of octahedra connected by corners. Two of the longest bonds in the GaO_6 and GeO_6 octahedra, GeO_2 (1.925 \AA) and GeO1 (2.057 \AA), are connected to a tetrahedral hole in the structure. The elongation of the bonds may indicate the presence of a hydrogen atom in this hole. Perhaps the hydrogen atom is not stable, but "migrates" between oxygen atoms in this tetrahedral hole.

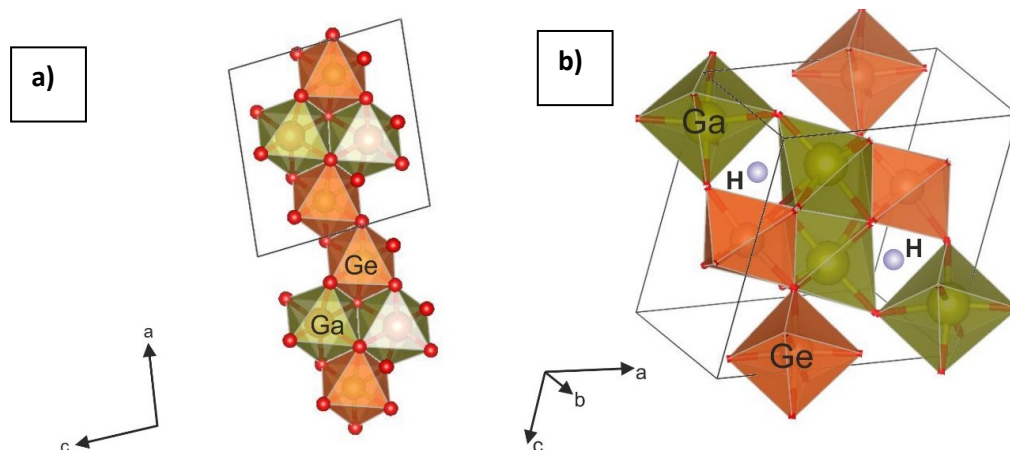


Fig. 2. Crystal structure of GaGeO_3OH . (a). The structural module in a - c projection. (b) The structure unit-cell and position of H atom in tetrahedral holes.

For the crystal structure of GaGeO_3OH ($P2_1/c$), the factor group analysis predicts $18A_g + 18B_g$ Raman active modes. The experimental Raman spectrum for the compound GaGeO_3OH was obtained in two regions of $150\text{--}1300\text{ cm}^{-1}$ and $2000\text{--}4000\text{ cm}^{-1}$ (Fig. 3). The experimentally obtained Raman spectrum has a topology similar to the Raman spectra of the Egg phase (natural and synthetic AlSiO_3OH and model MgSiO_4H_2). Bands in the low-frequency region of the spectrum $<300\text{ cm}^{-1}$ ($\sim 180, 190, 240, 267, 292$ and 311 cm^{-1}) can be attributed to vibrational and translational vibrational modes (Michel et al., 1996; Shi et al., 2020). Bands in the region $300\text{--}950\text{ cm}^{-1}$ ($\sim 339, 374, 387, 428, 477, 521, 603, 644, 701, 737, 826$ and 912) relate to vibrational modes of GeO_6 and GaO_6 octahedra (Rudolph et al., 2002; Rao et al., 2005; Shi et al., 2020). Several wide bands of $\sim 2330, 2436, 2878, 3152$, and 3430 cm^{-1} are recorded in the OH-group oscillation region. Such a different position of the OH-oscillation frequencies depends on both the length of the $\text{O}\cdots\text{O}$ and the angle H-O-O in the environment of the $\text{O-H}\cdots\text{O}$ bond (Hofmeister et al., 1999). Indeed, the calculated Raman spectra of the Mg-Egg phase suggest the presence of two vibrational modes at ~ 2850 and $\sim 3475\text{ cm}^{-1}$ corresponding to O-H bonds with different structural parameters (Solomatova et al., 2022). Thus, a group of wide bands recorded in the OH-group oscillation region indicates the presence of several positions of hydrogen in the crystal structure of GaGeO_3OH .

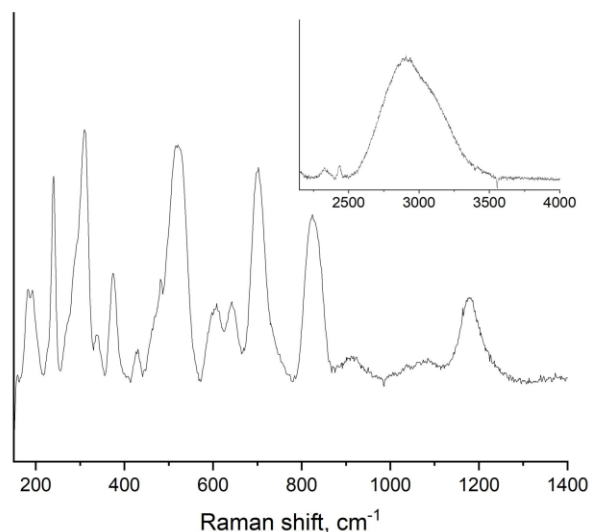


Fig. 3. The Raman spectra of the compound GaGeO_3OH .

Conclusions. For the first time, GaGeO_3OH crystals (up to $10\text{ }\mu\text{m}$) were synthesized at a temperature of 1000°C and a pressure of 7 GPa . The synthetic compound crystallizes in the same monoclinic structure with the space group $P2_1/n$ as the Egg phase AlSiO_3OH . The unit cell parameters of the synthesized GaGeO_3OH phase ($a = 7.5785(1)\text{ }\text{\AA}$,

$b = 4.4605(3)\text{ }\text{\AA}$, $c = 7.2469(4)\text{ }\text{\AA}$, $\beta = 97.519(2)^\circ$, $V = 242.87(02)\text{ }\text{\AA}^3$) are characterized by higher values compared to the natural and synthetic Egg phases. The factor group analysis predicts that 36 modes are very active. The experimentally obtained Raman spectrum has a topology similar to the Raman spectra of the Egg phase (natural and synthetic AlSiO_3OH and model MgSiO_4H_2). A group of wide OH-oscillation bands was detected in the high-frequency region of the Raman spectra, which indicates the presence of several hydrogen positions in the crystal structure of GaGeO_3OH . Thus, the synthetic compound GaGeO_3OH can be considered as an analog of the Egg phase.

Acknowledgments

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Iskrina A.V., Spivak A.V., Setkova T.V., Zakharchenko E.S., Viryus A.A. Comparative analysis of stability of GaGeO₃OH and AlSiO₃OH at high pressure (up to 30 GPa).

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Abstract: Gallogermanates have attracted sufficient attention as a high-pressure model of silicates and aluminosilicates due to chemical deformation of the structure. Thus, the GaGeO₃OH compound can be considered as an analog of the phase Egg (AlSiO₃OH), which is known as a possible H₂O reservoir in the deep Earth's interior. Using *in situ* Raman spectroscopy at high pressures, the dependence of the shift in the position of the main bands of the GaGeO₃OH Raman spectrum on the pressure was established. According to the data of Raman spectroscopy up to 30 GPa, it was revealed that the GaGeO₃OH undergoes the possible structural changes at 4 and 14 GPa.

Keywords: phase Egg, synthesis, gallium, germanium, high pressure, DAC, Raman spectroscopy

Introduction. Germanates and gallogermanates are an important class of materials attracting great interest due to a suitable combination of structural, optical, mechanical, electromagnetic and other properties for a wide range of applications in various devices (Belokoneva, 1994; Bu et al., 1998; Zhang et al., 2012; Luo et al., 2017; Lipina et al., 2019). Moreover, it was previously shown that germanates are model analogues of silicates of deep geospheres (Ringwood et al., 1963; Ringwood, 1970; Reid et al., 1970), which was confirmed in more modern works (Finger et al., 2000; Thomas et al., 2008; Stan et al.,

2017). On the other hand, gallium is a chemical analog of aluminum – their outer electron shells are identical to each other like a Si–Ge pair. This geochemical similarity of aluminum with gallium and silicon with germanium makes it possible to use the Ga₂O₃–GeO₂–H₂O system as a high-impact model of the Al₂O₃ – SiO₂ – H₂O system, in which most of the research was focused on studying the stability of the AlSiO₃OH Egg phase (spatial group (gr.) P2₁/n), first obtained by Eggleton in 1978 (Eggleton et al., 1978). Later, it was shown that the Egg phase is stable in the PT transition zone up to 21.5 GPa at 1500 °C (Fukuyama et al., 2017). *In situ* structural analysis at high pressure of the AlSiO₃OH Egg phase showed that up to 16 GPa, the *b* axis is the most compressible direction, and the β angle decreases (Schulze et al., 2018). *Ab initio*, a study of the structure and relative stability of MgSiO₄H₂ polymorphs predicted a new MgSiO₄H₂ phase (pr. gr. P4₃2₁2) with a stability field in the range of 0–14 GPa at 1500°C (Solomatova et al., 2022). The purpose of this work is to study the stability of a new compound, GaGeO₃OH, an analog of the Egg phase AlSiO₃OH, under baric exposure by *in situ* Raman spectroscopy.

Synthesis. Experiments on the synthesis of GaGeO₃OH were carried out using a toroidal anvil apparatus at P=7.0 GPa and T=1000°C (exposure time is 6 hours) at the D.S. Korzhinsky Institute of Experimental Mineralogy of the Russian Academy of Sciences (IEM RAS). A starting mixture of powders of gallium (99.5%) and germanium (99.9%) oxides in a 1:1 ratio was placed in a Pt ampoule with the addition of excess deionized water. The accuracy of pressure and temperature determination is ± 0.25 GPa and $\pm 20^\circ\text{C}$, respectively (Litvin, 1991). After the experiment, the sample was quenched to 20°C at a rate of $300 \pm 50^\circ\text{C/s}$.

SEM, EMPA. The composition of the obtained crystals was determined by electron-microprobe X-ray spectral analysis on a Tescan Vega II XMU scanning electron microscope using an energy-dispersive X-ray spectrometer (INCAx-sight) at an accelerating voltage of 20 kV, an absorbed electron current of 0.2 nA, and a probe size of 0.3 microns. The analysis time was 70 seconds. The following standards were used for the quantitative determination of the main elements: for Ge – Ge metal, for Ga – GaP.

X-ray diffraction analysis. X-ray diffraction analysis was performed by powder diffraction. The powder diffraction spectra were captured using a BRUKER Advance D8 instrument using a copper cathode X-ray tube and a capillary for examining small amounts of the sample. The resulting model of the structure was further refined by the Rietveld method in the TOPAS program to the confidence

factors R_{wp} of about 4% and R_{bragg} of 1.5%. The resulting structure corresponds to the well-known type $AlSiO_3OH$ (PhaseEGG).

Raman Spectroscopy at high pressures. Experiments on high-pressure Raman spectroscopy were carried out in a diamond anvil cell (DAC) at pressures up to 30 GPa. The DACs were equipped with anvils with a working surface size of 250 microns. The limiting gasket made of rhenium foil with a thickness of 200 microns was pre-pressed by about 30 microns, and then a hole with a diameter of ~ 125 microns was drilled into it. Several $GaGeO_3OH$ crystals were loaded into the resulting "chamber". NaCl was used as a pressure transfer medium. The use of NaCl is justified for pressures up to ~30 GPa (Mi, 2013). The pressure was determined by the calibrated shift of the fluorescence line of ruby R1 (a ruby sphere with a diameter of about 5 microns was placed inside the "chamber" of a cell with diamond anvils). The Raman spectra of $GaGeO_3OH$ crystals in a diamond anvil cell were captured on an installation consisting of an Acton SpectraPro-2500i spectrograph with a Pixis2K CCD detector cooled to -70 °C and an Olympus microscope with a continuous solid-state monomeric laser with a

wavelength of 532 nm. The laser beam was focused on the sample using an Olympus 50× lens into a spot with a diameter of ~5 μm. The signal accumulation time was 540s (3x180s). The software Fityk 1.3.1 was used for profile fitting for the Raman band analysis.

Results and discussion. The $GaGeO_3OH$ phase crystallizes in the $P2_1/n(14)$ space group with unit cell parameters: $a = 7.5785(1) \text{ \AA}$; $b = 4.4605(3) \text{ \AA}$; $c = 7.2469(4) \text{ \AA}$; $V = 242.87(02) \text{ \AA}^3$; $\beta = 97.519(2)^\circ$, while forming separate white crystals of isometric shape of pseudocubic habitus up to 5-8 microns in size (Spivak et al., 2025).

The factor group analysis for the crystal structure of $GaGeO_3OH$ ($P2_1/c$) suggests the following distribution of vibrational modes: IR active $18A_u + 18B_u$ ($\Gamma_{acoustic} = A_u + 2B_u$); Raman active $18A_g + 18B_g$. The experimental Raman spectrum for the $GaGeO_3OH$ compound was obtained in the range of $150\text{--}4000 \text{ cm}^{-1}$ (Fig. 1). The assignment of the main vibrational modes was performed by comparing our data with studies of similar compounds and the general position of the bands on the Raman spectra.

Table 1. Parameters of unit cells of gallium- and germanium-containing feldspars

Feldspar	a [Å]	b [Å]	c [Å]	α [°]	β [°]	γ [°]	V [Å ³]	n
NaAlGe ₃ O ₈	8.387(1)	13.292(2)	7.281(1)	94.28(1)	115.97(1)	90.66(1)	726.8(3)	58
KAlGe ₃ O ₈	8.854(2)	13.842(2)	7.315(2)	90.0	115.48(1)	90.0	809.3(3)	40
CaAl ₂ Ge ₃ O ₈	8.289(2)	12.695(2)	7.309(1)	93.43(2)	116.67(2)	93.31(1)	682.9(4)	41
CaAlGaSi ₂ O ₈	8.217(1)	12.940(1)	7.121(1)	93.02(1)	115.74(1)	91.29(1)	680.2(2)	70
BaAlGaSi ₂ O ₈	8.661(1)	13.105(1)	7.232(1)	90.0	114.57(1)	90.0	746.6(3)	60
Ca _{0.6} Sr _{0.4} AlGaSi ₂ O ₈	8.309(1)	13.014(2)	7.170(1)	91.90(1)	116.13(1)	91.44(1)	695.1(2)	56
Ca _{0.5} Sr _{0.5} AlGaSi ₂ O ₈	8.347(2)	13.015(2)	7.134(1)	91.95(1)	115.68(1)	91.09(1)	697.5(3)	49
CaAlGaSiGeO ₈	8.233(2)	13.456(1)	7.191(1)	93.10(1)	115.18(1)	91.79(1)	718.6(3)	54
SrAlGaSiGeO ₈	8.482(2)	13.367(2)	7.228(2)	90.0	115.11(1)	90.0	742.1(4)	37
BaAlGaSiGeO ₈	8.746(2)	13.266(2)	7.259(2)	90.0	114.85(1)	90.0	764.3(4)	52

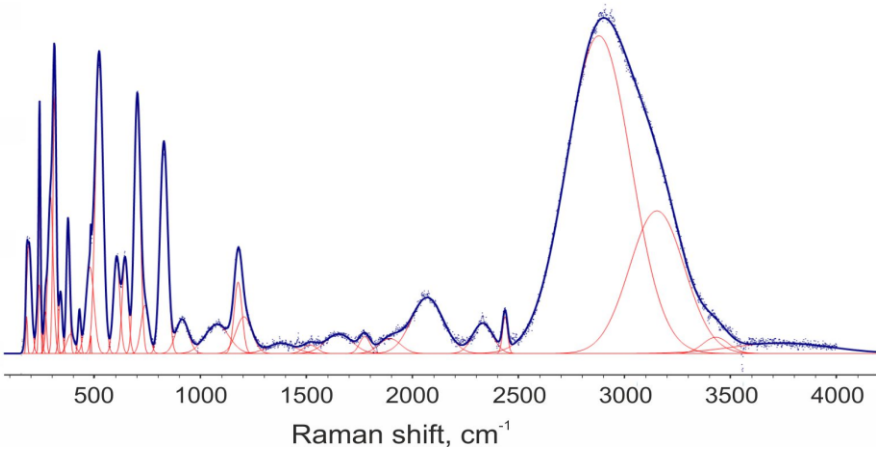


Fig. 1. The Raman spectra of the compound $GaGeO_3OH$.

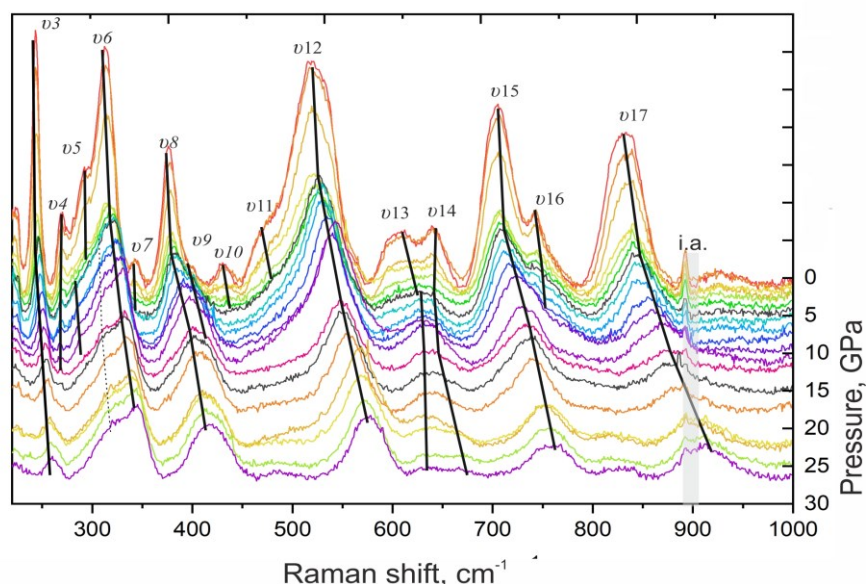


Fig. 2. Evolution of the Raman spectrum of GaGeO_3OH in pressure up to ~ 30 GPa. Grey area is an instrumental artefact (i.a.) of the DAC.

A detailed study of the Raman spectra of GaGeO_3OH up to ~ 30 GPa traced the evolution of 15 Raman bands (ν_3 - ν_{17}) under compression (Fig. 2, Table 1). For bands in the region of OH-group oscillations, frequency trends could not be determined, since the intensities of these bands turned out to be relatively small during compression and decompression. In general, all the Raman bands shift to the region of higher wave numbers and become less intense with increasing pressure. As a result of processing the obtained spectroscopic data, changes in the coefficients of dependence of the frequency shifts of the Raman modes with increasing pressure were revealed. Based on this, three pressure ranges were identified depending on the value of $\partial\nu/\partial P$ ($\text{cm}^{-1}/\text{GPa}$) (Table 2). The trends in the wavenumber values for most bands in the first pressure range (from atmospheric pressure to ~ 4 -5 GPa) are characterized by an insignificant value of $\partial\nu/\partial P = 0.24$ - 1.23 $\text{cm}^{-1}/\text{GPa}$. At the same time, the bands of valence vibration GeO_6 (ν_{17} and ν_{13}) and

the band in the region of deformation vibrations and/or bending vibrations GaO_6 (ν_{11}) have the highest value $\partial\nu/\partial P = 3.98$ (0.28), 3.29 (0.18) and 3.09 (0.51) $\text{cm}^{-1}/\text{GPa}$, respectively. It should be noted that ν_{14} and ν_9 have negative pressure dependence coefficients. Some less intense bands ν_7 , ν_{10} , and ν_{11} disappear at ~ 4 -5 GPa at the end of the first pressure region (Fig. 2). The coefficients of change in the wavenumber values of most bands in the second pressure region (from ~ 4 -5 to ~ 14 GPa) have a higher value $\partial\nu/\partial P = 0.51$ - 2.82 $\text{cm}^{-1}/\text{GPa}$ compared to the first region. This indicates that the compression of bonds occurs more intensively in the second pressure region. The valence band GeO_6 (ν_{17}) still has the highest value $\partial\nu/\partial P = 3.89$ (0.15) $\text{cm}^{-1}/\text{GPa}$. The remaining "shoulder" bands ν_4 , ν_5 , ν_9 , and ν_{16} disappear at ~ 14 GPa at the end of the second pressure region (Fig. 2). The disappearance of the bands in the Raman spectrum may be due to a decrease in intensity against the general background with increasing pressure.

Table 1. The comparison of the band position of Raman spectra of AlSiO_3OH , MgSiO_4H_2 and GaGeO_3OH at ambient conditions.

The assignment of the vibrational modes	$\nu\#$	Ga,Ge-Egg GaGeO_3OH synthesized at 7 GPa/1000 °C This work	Phase Egg AlSiO_3OH synthesized at 17 GPa/1000 °C (Xue et al., 2006)	Phase Egg AlSiO_3OH Inclusions in diamond (Wirth et al., 2007)	Mg-Egg MgSiO_4H_2 <i>Ab initio</i> (Solomatova et al., 2022)
O-Ga-O and/or O-Ge-O libration and Ga-O and/or Ge-O translation (Shi et al., 2020; Michel et al., 1996)	ν_1	179.53	255		
	ν_2	190.21			
	ν_3	240.09			
	ν_4	267.18			
	ν_5	291.76			
	ν_6	311.28			
GaO_6 deformation and bending (Shi et al., 2020; Rao et al., 2005; Rudolph	ν_7	338.82	374	360	
	ν_8	374.19			
	ν_9	386.86			

The assignment of the vibrational modes	ν #	Ga,Ge-Egg GaGeO ₃ OH synthesized at 7 GPa/1000 °C This work	Phase Egg AlSiO ₃ OH synthesized at 17 GPa/1000 °C (Xue et al., 2006)	Phase Egg AlSiO ₃ OH Inclusions in diamond (Wirth et al., 2007)	Mg-Egg MgSiO ₄ H ₂ <i>Ab initio</i> (Solomatova et al., 2022)
et al., 2002)	ν 10	427.95	418	419	
	ν 11	477.50	474	478	
	ν 12	521.58	521	515	
GeO ₆ stretching (Okada et al., 2008; Verweij et al., 1979).	ν 13	603.33	610		
	ν 14	643.78			
	ν 15	700.64	754	757	
	ν 16	737.08			
	ν 17	826.34	898	868	
	ν 18	912.11	940		
	ν 19	1076.29			
OH bending libration (Xue et al., 2006)	ν 20	1176.77	1232		
	ν 21	1204.81			
	ν 22	2068.82	2150	2155	
OH stretching (Solomatova et al., 2022; Xue et al., 2006)	ν 23	2330.40			
	ν 24	2435.80	2450	2470	
	ν 25	2878.26	2800	2558	2850
	ν 26	3152.52		2695	
	ν 27	3430.69			3475

In the third pressure region (from ~14 to ~30 GPa), we observed eight remaining bands: ν 3, ν 6 in the area of vibrational O-Ga-O and/or O-Ge-O and translational Ga-O and/or Ge-O vibrations; ν 8, ν 12 in the area of deformation vibrations and/or bending vibrations of GaO₆; ν 13, ν 14, ν 15, ν 17 in the region of valence vibrations of GeO₆. Most of the bands in the third pressure region are characterized by a lower value of the coefficients of change in the wavenumber values than in the second region. After decompression, all intense bands (ν 3, ν 6, ν 8, ν 12, ν 15, and ν 17) returned to the initial position of the Raman shift. Also, only minor changes were observed in the Raman spectra of GaGeO₃OH compared with the initial spectra under standard conditions. In the Raman spectra of the sample "after high pressure", there is a decrease in the intensity of all bands and an almost complete disappearance of the "shoulder bands" (ν 7, ν 9, ν 18 and ν 19) compared with the initial sample "before high pressure". This phenomenon may be related to the amorphization of the sample under the influence of pressure, a similar phenomenon has been observed when pressure is applied to other compounds (e.g. Ruiz-Fuertes et al., 2016). Previous work on the stability of the structural characteristics of the AlSiO₃OH phase at high pressures showed that the b axis is the most compressible direction, and the β angle decreases to 16 GPa (Schulze et al., 2018). This is due to the fact that the length of the Si-O4 bond decreases, strengthening it up to a pressure of 16 GPa and at the same time weakening the O4-H bond. In our case, we observed the same effect for the GeO₆ (ν 17) valence

band; in all three pressure regions, this band has the highest value of $\partial\nu/\partial P$, which indicates a stronger compression of the Ge...O bond length compared to the others.

Table 2. Frequency shifts ($\partial\nu/\partial P$, cm⁻¹/GPa) of Raman bands of GaGeO₃OH.

ν #	Frequency shifts		
	$\partial\nu/\partial P$, cm ⁻¹ /GPa		
	I	II	III
ν 3	0.77 (0.06)	0.81 (0.06)	0.42 (0.01)
ν 4*	0.24 (0.03)	-0.04 (0.05)	Not observed
ν 5*	0.82 (0.16)	1.26 (0.10)	Not observed
ν 6	0.62 (0.17)	1.51 (0.12)	0.97 (0.06)
ν 7*	1.18 (0.06)	Not observed	Not observed
ν 8	1.23 (0.12)	2.06 (0.04)	1.39 (0.08)
ν 9*	-0.07 (0.12)	1.70 (0.04)	Not observed
ν 10*	1.14 (0.34)	Not observed	Not observed
ν 11*	3.09 (0.51)	Not observed	Not observed
ν 12	1.10 (0.16)	2.31 (0.09)	2.37 (0.05)
ν 13	3.29 (0.18)	0.80 (0.08)	0.50 (0.02)
ν 14	-0.88 (0.28)	0.51 (0.06)	2.10 (0.08)
ν 15	0.65 (0.20)	2.82 (0.14)	1.98 (0.04)
ν 16*	1.10 (0.08)	0.88 (0.41)	Not observed
ν 17	3.98 (0.28)	3.89 (0.15)	2.06(0.24)

* - shoulder bands.

Conclusions. *In situ* high pressures Raman spectroscopic study of a new GaGeO₃OH phase (an analog of the Egg AlSiO₃OH phase) up to ~ 30 GPa was made. A change in the spectral patterns of the Raman with a change in the values of the coefficient ($\partial\nu/\partial P$), accompanied by the disappearance of several bands at pressures of ~ 4 GPa and ~ 14 GPa, indicates possible structural transformations. The

band corresponding to the GeO₆ valence oscillation is characterized by the highest coefficient ($\partial\nu/\partial P$), which indicates the most compressible Ge···O bond in the entire pressure range up to ~ 30 GPa.

Acknowledgments

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Kotelnikov A.R., Suk N.I., Kotelnikova Z.A., Akhmedzhanova G.M., Drozhzhina N.A. Feldspars and their solid solutions: an experimental study. UDC 550.89:549.07

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Abstract. At a temperature of 500–900°C and a pressure of 1–3 kbar, solid solutions of feldspars of various compositions and their cation-exchange reactions with solutions were studied. Based on experimental data, Margules parameters were calculated to describe excess mixing energies of solid solutions and unit cell parameters were refined. A comparison with previously studied feldspars was made. Empirical dependencies for calculating the energy parameters of the Margules model and unit cell volumes for various minerals with a feldspar structure were proposed.

Keywords: *feldspars, synthesis, solid solutions, thermodynamic functions, crystallochemical parameters*

The reactions of cation exchange of solid solutions of feldspars with water-salt fluids were studied experimentally at temperature of 500–900 °C and pressure of 1–3 kbar. Feldspars with isomorphic substitutions of such elements as Na, K, Rb, Ca, Sr, Ba, B, Al, Ga, Fe^{3+} , Si, Ge, P, As were studied. In addition, a method of synthesis of solid solutions of feldspars at different temperatures was used to construct phase diagrams. The methods and results of the experiments are presented in the work (Kotelnikov, 1995). Excess mixing functions of solid solutions of feldspars were calculated based on original (Kotelnikov, 1995; Kroll H., et al., 1995) and literary data. Data on the values of excess mixing functions of solid solutions allow for comparison for a wide range of the studied feldspars and feldspathoids. The magnitudes of deviation from ideality can be conveniently estimated using the integrated values of G^e and V^e (Kotelnikov, 1995), which are obtained by integrating the values of the redundant functions:

$$G_{\text{int}}^e = \int_0^1 G^e dX_i; \quad (1)$$

$$V_{\text{int}}^e = \int_0^1 V^e dX_i; \quad (2)$$

where X_i – mole fraction of the i -th mineral of feldspars.

These integrated values allow comparison of different solid solutions of framework aluminosilicates regardless of the type of isomorphic substitutions. At the same time, the G_{int}^e value can only be used to compare the total deviation of solid solutions from ideality. For thermodynamic calculations, the values of the Margules model parameters, or numerical values of the activity coefficients, are used. To compare deviations from ideality of different solid solutions, these values (W^G_1 , W^G_2) had to be reduced to the same temperature. This was done by generalizing the literature data of various researchers. Table 1 shows the coefficients of the polynomials of the dependences of the Margules parameters on temperature: $W^G_{1,2} = A + B \times T + C \times T^2$. It was previously shown (Kotelnikov et al., 2021) that the values ($\langle T\text{-O} \rangle$) and R_i^T for feldspars are in a linear relationship:

$$\langle T\text{-O} \rangle = 1.2596 + 0.89324 \times (R_i^T) [\text{\AA}]. \quad (3)$$

The R_i^T value is the average size of a cation in a tetrahedral position of the feldspar crystal structure. Therefore, it is possible to calculate the average $\langle T\text{-O} \rangle$ bond lengths for feldspar structures based on the average size of a cation in a tetrahedral position. To evaluate the properties of feldspar solid solutions, it is convenient to use the parameter $A3 = \Delta R_i \times [\langle T\text{-O} \rangle]^3$, (Kotelnikov, 1995).

Since the G_{int}^e values for all studied feldspar solid solutions have positive values regardless of the isomorphism type, the absolute values of $A3$ were used to find the dependencies. For solid solutions with anorthite structure (Ca-Sr; Sr-Ba and Ca-Ba feldspars), the value $A3 = 2 \times (\Delta R_i) \times [R_i^T]^3$, which is due to the doubling of the lattice parameter of the anorthite type.

The dependence of the G_{int}^e values (750°C) on the $A3$ value is described by the following regression equation:

$$G_{\text{int}}^e = -0.1309 + 1.8508 \times A3; \quad S_x = 0.36; \quad V_x = 17\%; \quad E_x = 0.17; \quad E_x(\text{relative}) = 8\% \quad (4)$$

For the parameters of the Margules model (at 750°C), the following dependencies on the value of $A3$ were calculated:

$$W^G_1 = -0.2230 + 9.6265 \times A3; \quad S_x = 3.2; \quad V_x = 28\%; \quad E_x = 1.6; \quad E_x(\text{relative}) = 14\% \quad (5)$$

$$W^G_2 = -2.246 + 13.1489 \times A3; \quad S_x = 3.3; \quad V_x = 24\%; \quad E_x = 1.6; \quad E_x(\text{relative}) = 12\% \quad (6)$$

Thus, there is a clear connection between the values of excess energies and crystallochemical parameters. The values of the radii of cations in the tetrahedral position and isomorphic cations allow us to estimate the degree of non-ideality of solid solutions of various feldspars in the temperature range of 500-850°C.

Conclusions

1. For various solid solutions of feldspar, the temperature dependences of the Margules coefficients were calculated to describe the excess mixing energies of the solid solutions.

2. The quantitative relationship between the Margules parameters (W^G_1 and W^G_2) and the crystal-chemical parameters of the feldspar solid solutions was shown.

Table 1. Values of the polynomial coefficients for calculating the Margules parameters (of the type $W^G_{1,2} = a + b \times TK + c \times (TK)^2$) of solid solutions of feldspars

System Fsp	Isomorphism	a (W^G_1)	b	c	a (W^G_2)	b	c
Na-Ca-Al-Si	Na-Ca	89.7000	-0.14300	0.000060	84.1800	-0.14250	0.000060
Na-Ca-Ga-Si	Na-Ca	-1.4750	0.00106	-	10.6370	-0.00290	-
Na-Sr-Al-Si	Na-Sr	20.8380	-0.01340	-	27.4050	-0.01760	-
Na-Ba-Al-Si	Na-Ba	15.6300	-0.00512	-	22.0500	-0.00855	-
K-Ca-Al-Si	K-Ca	65.023	-0.0397	-	51.902	-0.0345	-
K-Sr-Al-Si	K-Sr	31.0600	-0.01860	-	40.2500	-0.02960	-
K-Ba-Al-Si	K-Ba	28.8100	-0.02380	-	49.5100	-0.04461	-
Rb-Ba-Al-Si	Rb-Ba	8.870	-0.00305	-	12.2060	-0.00460	-
Ca-Sr-Al-Si	Ca-Sr	60.0360	-0.04192	-	59.7260	-0.04169	-
Sr-Ba-Al-Si	Sr-Ba	26.4190	-0.01601	-	26.2810	-0.01590	-
Ca-Ba-Al-Si	Ca-Ba	81.0700	-0.05500	-	81.8200	-0.04370	-
Na-K-Al-Si	Na-K	21.1000	-0.01010	-	31.0500	-0.01380	-
Na-K-Ga-Si	Na-K	19.6000	-0.00875	-	31.8600	-0.01399	-
Na-Rb-Al-Si	Na-Rb	110.8000	-0.09170	-	148.9000	-0.11850	-
K-Rb-Ga-Si	K-Rb	7.1390	-0.00345	-	6.8640	-0.00320	-
Na-K-Al-Ge	Na-K	30.7200	-0.01400	-	30.9400	-0.00600	-
Na-B-Al-Si	B-Al	66.1370	-0.05464	-	38.4600	-0.02828	-
Ca-Ba-Al-Ge	Ca-Ba	65.0820	-0.04400	-	44.5230	-0.02535	-
K-Rb-Al-Si	K-Rb	9.9950	-0.00630	-	9.9950	-0.00630	-

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Kotelnikov A.R., Korneeva A.A., Kosova S.A., Suk N.I., Drozhzhina N.A. Synthesis of feldspars with gallium and germanium. UDC 550.89:549.07

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Abstract. Feldspars containing gallium and germanium were synthesized by pyrosynthesis (at 1300°C) and hydrothermal recrystallization (at 650°C and P=2 kbar). Experiments on cation exchange equilibria with 1M SrCl₂ solution were conducted at 750 and 650°C (P=2 kbar). Their solid solutions were synthesized. The unit cell parameters of the synthesized feldspars and their solid solutions were clarified. These feldspars were synthesized for the first time.

Keywords: synthesis, feldspars, gallium, germanium, solid solutions

Gallium- and germanium-containing feldspars were synthesized from stoichiometric mixtures of carbonates and oxides, pre-annealed at 1300°C for 12 hours. Then, hydrothermal synthesis of feldspars was carried out from pyrosynthetic material at 650 and 750°C and a pressure of 2 and 3 kbar. Pyrosynthetic samples were placed in platinum (or gold) ampoules, into which a small amount of water (~10 wt.%) was

poured. The duration of the experiments was 12-21 days. After the experiments, the samples were studied using microprobe and X-ray analysis.

The compositions of the synthesized solid products of the experiments were determined by the method of local X-ray spectral microanalysis using a Tescan Vega II XMU scanning electron microscope (Tescan, Czech Republic) equipped with an INCA Energy 450 X-ray spectral microanalysis system with energy-dispersive (INCAx-sight) and crystal-diffraction (INCA wave 700) X-ray spectrometers

(Oxford Instruments, England) and the INCA Energy+ software platform. The analysis conditions using only an energy-dispersive spectrometer were as follows: accelerating voltage of 20 kV, current of absorbed electrons on Co of 0.3 nA, analysis time at a point of 70 s. X-ray study of solid solutions of gallium-containing alkali feldspars was carried out on an HZG-4 and “Bruker” diffractometer in the continuous scanning mode. Spectral-purity silicon ($a=5.4307$ [Å]) was used as an internal standard.

Table 1. Parameters of unit cells of gallium- and germanium-containing feldspars

Feldspar	a [Å]	b [Å]	c [Å]	α [°]	β [°]	γ [°]	V [Å ³]	n
NaAlGe ₃ O ₈	8.387(1)	13.292(2)	7.281(1)	94.28(1)	115.97(1)	90.66(1)	726.8(3)	58
KAlGe ₃ O ₈	8.854(2)	13.842(2)	7.315(2)	90.0	115.48(1)	90.0	809.3(3)	40
CaAl ₂ Ge ₃ O ₈	8.289(2)	12.695(2)	7.309(1)	93.43(2)	116.67(2)	93.31(1)	682.9(4)	41
CaAlGaSi ₂ O ₈	8.217(1)	12.940(1)	7.121(1)	93.02(1)	115.74(1)	91.29(1)	680.2(2)	70
BaAlGaSi ₂ O ₈	8.661(1)	13.105(1)	7.232(1)	90.0	114.57(1)	90.0	746.6(3)	60
Ca _{0.6} Sr _{0.4} AlGaSi ₂ O ₈	8.309(1)	13.014(2)	7.170(1)	91.90(1)	116.13(1)	91.44(1)	695.1(2)	56
Ca _{0.5} Sr _{0.5} AlGaSi ₂ O ₈	8.347(2)	13.015(2)	7.134(1)	91.95(1)	115.68(1)	91.09(1)	697.5(3)	49
CaAlGaSiGeO ₈	8.233(2)	13.456(1)	7.191(1)	93.10(1)	115.18(1)	91.79(1)	718.6(3)	54
SrAlGaSiGeO ₈	8.482(2)	13.367(2)	7.228(2)	90.0	115.11(1)	90.0	742.1(4)	37
BaAlGaSiGeO ₈	8.746(2)	13.266(2)	7.259(2)	90.0	114.85(1)	90.0	764.3(4)	52

The polygonal method of correction of X-ray reflections was used (Kroll et.al., 1995; Kotelnikov, 1995). The obtained results allowed us to calculate the unit cell parameters of solid solutions. The unit cell parameters (UCP) were refined using 45–103 reflections in the angular range of 6.5–44.5° (Θ). The parameters were calculated using the LCC, PUDI, and MINCRYST programs (Burnham, 1991; Chichagov, 1994). The UCP values are given in Table 1.

In addition to hydrothermal synthesis, the method of cation exchange equilibria was used: $\text{CaAlGaSi}_2\text{O}_8 + \text{SrCl}_2(\text{aq}) = \text{SrAlGaSi}_2\text{O}_8 + \text{CaCl}_2(\text{aq})$. This method allowed us to obtain solid solutions of feldspars with the replacement of calcium by strontium. The UCP of the solid solutions are also given in Table 1. The data obtained are useful for studying the crystallochemical features of solid solutions of framework aluminosilicates and developing ideas about the geochemistry of rare elements in the earth's crust.

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Zircono- and titanosilicates - indicators of alkalinity in alkaline pegmatites (by experimental data).

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Abstract. At temperature 600 °C and pressure 2 kbar, zircon- and titanosilicates were synthesized in the presence of fluids of varying alkalinity. Such parageneses are observed in postmagmatic formations of the Khibiny and Lovozero ultra-agpaitic complexes. The duration of the experiments was 10 days. The following mineral-like

phases were diagnosed in the experimental products using electron probe microanalysis and X-ray fluorescence: minerals of the eudialyte group, minerals of the lovozerite group, parakeldyshite, occurring in ultra-alkaline postmagmatic formations.

Keywords: *synthesis, zirconosilicates, eudialyte, lovozerite, lowenite, postmagmatic processes, alkaline massifs, alkaline fluid.*

Previously, in order to obtain information on the conditions of zirconium and titanate formation, as well as to study the effect of the fluid regime on the composition of the forming phases, a team of authors conducted a series of experiments on the synthesis of zirconium and titanate silicates in alkaline conditions under various physicochemical conditions (solutions of NaCl, NaF, NaOH, Na₂CO₃ of various concentrations were used as fluids) (Kovalskaya et al., 2023; Kovalskaya et al., 2024). The experiments were carried out in hermetically sealed platinum ampoules on a high gas pressure unit (HGU) and hydrothermal units (UVD-10000) designed by the IEM RAS, at a temperature of 500 - 600 °C and a pressure of 2 kbar. The sample: solution ratio was 10:1. The duration of the experiments varied from 10 to 21 days. The experimental products were studied by electron probe microanalysis, X-ray diffraction and IR spectroscopy. The results of the experiments indicate the influence of the fluid regime on the composition of the crystallizing phases under the same conditions of temperature and pressure. During crystallization of zirconium silicates in a less alkaline solution (1M NaF, 1M NaCl with 0-10% NaOH), growth of zirconium silicates (eudialyte, parakeldyshite, lovenite and its Fe analogue, burpalite, zircon) is observed, and in a more alkaline solution (1M NaF with 20% NaOH), lueshite and fluoride (F analogue of lakargite CaZrF₆) are formed. In the synthesis of titanate silicates, the

addition of 15% NaOH and 10% Na₂CO₃ to 1M NaCl does not significantly affect the crystallization of mineral phases (joint growth of lovenite and lueshite is observed everywhere with some variations of other associated phases - titanite, pectolite and serandite in the first case, titanite in the second case and schisolate, natisite (in the third). Recently, experiments were carried out on the synthesis of zirsinalite. Zirsinalite Na₆(Ca,Mn,Fe)ZrSi₆O₁₈, was discovered in the Khibiny alkaline massif (Fig. 1), and was found in several points of the Lovozero massif. Zirsinalite from Mount Karnasurt has been studied in most detail, where it forms colorless rims and pseudomorphs on eudialyte and, in turn, is replaced by lovozerite; it associates with natrosilite, aegirine, analcime, sodalite, nepheline, feldspar, cancrinite, natrolite, villiaumite, thermanatrite, lomonosovite, woonnemite, kazakovite, etc. (Pekov, 2001).



Fig. 1. Tsirsinalite (pseudomorphosis on eudialyte) in association with lovozerite, microcline, aegirine, Yukspor mountain, Khibiny alkaline massif, Kola Peninsula.

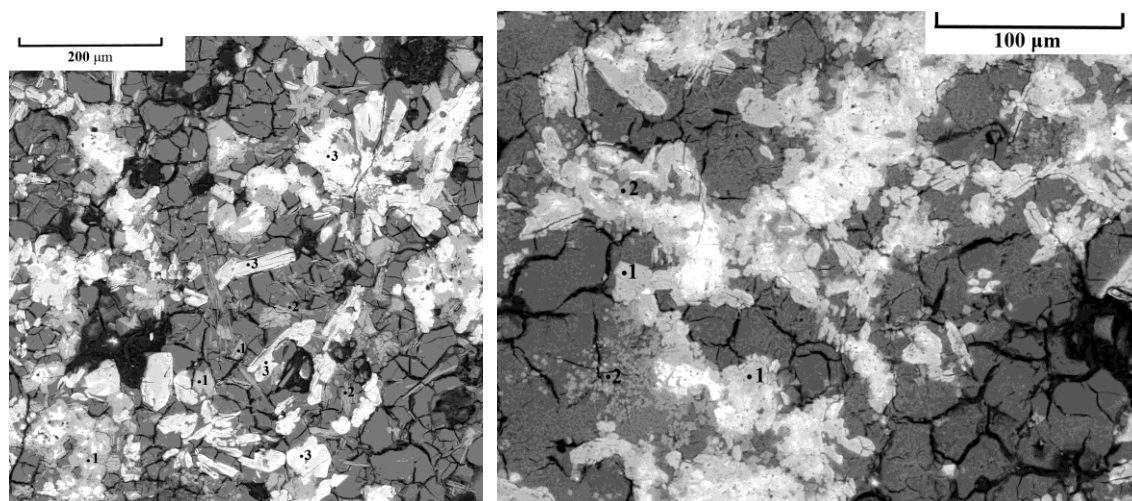


Fig. 2. Results of synthesis of experiment 87. 1 – tsirsinalite, 2 – pectolite, 3 – vlasovite. Back-scattered electron image.

Fig. 3. Results of synthesis of experiment 88. 1 – vlasovite, 2 – pectolite. Back-scattered electron image.

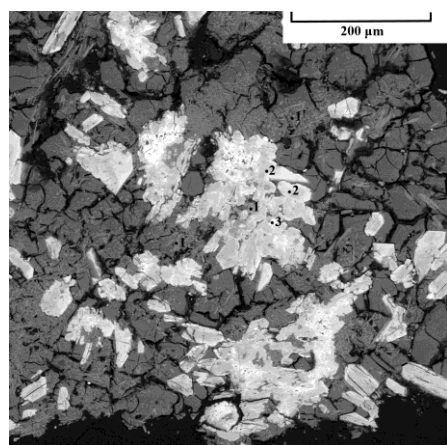


Fig. 4. Results of the synthesis of experiment 89. 1 – pectolite, 2 – vlasovite, 3 – parakeldyshite. Back-scattered electron image.

Table 1. Chemical composition of phases obtained during experiments 87-89.

Compo- nents	Exp. 87			Exp. 88		Exp. 89		
	Vlasovite	Tsirsina- lite	Pectolite	Vlaso- vite	Decationized pectolite with Zr impurity	Vlasovite	Para- keldyshite	Pectolite
weir. %								
Na ₂ O	12.39	12.51	9.61	12.46	13.76	12.61	17.34	9.80
CaO	bdl	12.56	33.00	bdl	14.67	0.34	0.34	31.99
ZrO ₂	29.674	17.12	1.09	29.17	14.71	31.09	42.62	bdl
HfO ₂	2.16	bdl	bdl	bdl	bdl	bdl	bdl	bdl
SiO ₂	55.68	55.45	55.47	56.98	56.40	55.86	37.74	56.67
Nb ₂ O ₅	bdl	bdl	bdl	bdl	bdl	1.88	bdl	bdl
Σ	99.89	97.65	99.16	98.62	99.53	101.78	98.05	98.46
formula coefficients								
Na	1.73	2.63	1.01	1.70	1.42	1.75	1.78	1.00
Ca	0	1.46	1.91	0	0.84	0.03	0.02	1.81
Zr	1.04	0.90	0.03	1.00	0.38	1.08	1.10	0
Hf	0.04	0	0	0	0	0	0	0
Si	4.00	6.00	3.00	4.00	3.00	4.00	2.00	3.00
Nb	0	0	0	0	0	0.06	0	0

The experiments were carried out from a sol-gel of the zirsinalite composition using 10, 20 and 30% NaOH solutions. The experiments were carried out on a hydrothermal setup (UVD-10000) designed by the IEM RAS at a temperature of 550 °C and a pressure of 2 kbar. In experiment 87 with a 10% NaOH solution, vlasovite, decationized zirsinalite and pectolite were formed; in experiment 88 with a 20% NaOH solution, vlasovite and pectolite were formed, and in experiment 89 with a 30% NaOH solution, vlasovite, parakeldyshite and pectolite were formed (Table 1, Figures 2-4).

The infrared spectroscopic study was performed in accordance with the State Assignment theme, state record number AAAA-A19-119092390076-7, and the rest of the work was performed under State Assignment theme FMUF-2022-0002

Kuzyura A.V., Spivak A.V., Setkova T.V., Iskrina A.V., Zakharchenko E.S., Viryus A. A. High-pressure synthesis of Ge-analogues of olivine (α -Mg₂GeO₄) and spinel (γ -Mg₂GeO₄). UDC 550.4.02, 549.057

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Abstract. Geochemical similarity of silicon to germanium allows using germanates as analogues of planetary silicates. In addition, as a rule, germanates undergo similar phase transitions, but at significantly lower pressures than similar silicate minerals. In this regard, a test high-pressure synthesis of Mg-germanates were carried out. Crystals of α -Mg₂GeO₄ olivine with size up to 100 μm were obtained in the Mg(OH)₂-MgCO₃-GeO₂ system at pressure of 5 GPa and temperature of 900 °C. Crystals of γ -Mg₂GeO₄ spinel with size up to 50 μm were synthesized in the MgO-GeO₂-H₂O system at 7 GPa and 900 °C. Phase identification was carried out using electron probe analysis and Raman spectroscopy.

Keywords: HPHT experiments; synthesis; germanium; Ge-analogues of minerals; Raman-spectroscopy

The geochemical similarity of silicon to germanium allows the use of germanates as analogues of planetary silicates (Dutta et al., 2018; Finger and Hazen, 2000; Reid and Ringwood, 1970; Ringwood, 1970; Stan et al., 2017). Olivine (Mg,Fe)₂SiO₄ is the most abundant mineral in the

Earth's upper mantle. A cause of the main seismic boundaries (410 and 520 km) in the upper mantle and the transition zone is the phase transitions of the Mg-rich olivine α -(Mg,Fe)₂SiO₄ into wadsleyite β -(Mg,Fe)₂SiO₄ and ringwoodite γ -(Mg,Fe)₂SiO₄ (Ringwood 1991). Synthetic germanium analogues of olivine and spinel α,γ -X₂GeO₄ (X=Mg, Mn, Co....) and their phase transformations have been studied since the 70s of the last century and has not lost its relevance today (Akimoto, 1970; Divya et al., 2024; Dupas-Bruzek et al., 1998; Ross et al., 1987; Shi et al., 2015). Currently solid-phase synthesis is the most available methods of synthesis of Ge-analog for subsequent researches. Olivine (α -Mg₂GeO₄) was synthesized from magnesium and germanium oxides at 1473-1770 K (1-5 days) (Divya et al., 2024; Fiquet et al., 1992; Ross et al., 1987). Spinel (γ -Mg₂GeO₄) can be synthesized at 0.07 GPa and 973-1048 K (5 days) by the hydrothermal method using previously obtained olivine as a starting material (Ross et al., 1987). A high-impact and high-temperature experiment was used to study the phase state of Mn₂GeO₄. The α -modification was synthesized at 2.5-4 GPa, β - at 4-6 GPa, and δ - at 6-7 GPa in the temperature range of 800-1200 °C (Akimoto, 1970). In addition, the kinetic characteristics of the α -Mg₂GeO₄→ γ -Mg₂GeO₄ phase transition were estimated at pressure range 0-50 kbar and 25-1100 °C (Lauterjung and Will, 1986). A phase diagram with generalized data on the α -Mg₂GeO₄→ γ -Mg₂GeO₄ phase transition at that time was presented by Ross et al. at a pressure up to 4 GPa and a temperature up to 1900 °C (Ross et al., 1987). In this paper, we give results for high-pressure (5 and 7 GPa) and high-temperature synthesis of two modifications: olivine α -Mg₂SiO₄ and spinel γ -Mg₂GeO₄ for their subsequent using at experimental researches at high pressure.

The synthesis experiments were carried out with using of toroidal anvils press at the D.S. Korzhinsky Institute of Experimental Mineralogy of the Russian Academy of Sciences (IEM RAS) (Litvin, 1991). Magnesium oxides (99.9%), germanium oxide (99.9%), magnesium hydroxide (99.5%) and magnesium carbonate (99.5%) were used as the starting material. Starting sets within the model systems Mg(OH)₂-MgCO₃-GeO₂ and MgO-GeO₂-H₂O (deionized water) were placed in an Au ampoule. The experiments were carried out at P=5.0 and 7.0 GPa and T=900 °C. The pressure was determined using the standard Bi polymorphism (at 2.55, 2.7, and 7.7 GPa) with an accuracy of ± 0.25 GPa. The temperature was measured and controlled by a thermocouple (Pt₇₀Rh₃₀/Pt₉₄Rh₆) with an accuracy of ± 20 °C. After the experiment, the sample was cooled upto 20 °C at a rate of 300 ± 50

°C/sec.

The habitus of the obtained crystals, surface structure, and composition were studied using the Tescan Vega II XMU scanning electron microscope (SEM) with secondary and reflected electron detectors, as well as an energy-dispersive X-ray spectrometer (INCAx-sight) for quantitative analysis. Electron probe X-ray spectral analysis of the studied samples was performed at an accelerating voltage of 20 kV, an absorbed electron current of 0.2 nA, and a probe size of 0.3 microns. The analysis time was 70 seconds. The following standards were used for the quantitative determination of the major elements: for Ge – Ge elementary, for Ga – GaP.

The Raman spectra of the crystals were captured on a setup consisting of an Acton SpectraPro-2500i spectrograph with a Pixis2K CCD detector cooled to -70 °C and an Olympus microscope with a continuous solid-state single-mode laser with a wavelength of 532 nm. The laser beam was focused on the sample using an Olympus 50× lens into a spot with a diameter of ~5 µm. The signal accumulation time was 540 sec (3x180sec). The obtained spectra were processed using of the Fityk 1.3.1 program.

Results of experiments in the system (Mg(OH)₂-MgCO₃)₅₀-(GeO₂)₅₀ are represented by the following phases: olivine α -Mg₂SiO₄, spinel γ -Mg₂GeO₄, and magnesite MgCO₃ (Fig.1a). The chemical composition of the obtained phases corresponds to the crystallochemical formulas of these minerals. Ge-olivine forms prismatic crystals up to 100 microns in size with inclusions of Ge-spinel and magnesite. Small (first microns) octahedral single crystals and Ge-spinel twins form local groups in the total fine crystalline mass of magnesium carbonate (fig. 2a). According to Raman spectroscopy data, residual Mg(OH)₂ was not detected.

The obtained result on the joint formation of germanium phases α -Mg₂SiO₄ and γ -Mg₂GeO₄ at pressure of 5 GPa and temperature of 900 °C is consistent with data on the kinetics of transformation α -Mg₂GeO₄→ γ -Mg₂GeO₄ at 50 kbar and 1100 °C (Lauterjung and Will, 1986).

In the case of using of the MgO-SiO₂ starting system with an excess of H₂O (deionized water) at pressure of 7 GPa and temperature of 900 °C, we have obtained a white fine crystalline powder consisting of crystals of cubic habitus. They are well faceted, octahedral in shape. Twins are often occurs among them, the size of the individuals is up to 20-30 microns (Fig. 1 b). The chemical composition of the resulting phase corresponds to the crystallochemical formula Mg₂GeO₄. According to Raman spectroscopy data, the modification of spinel γ -Mg₂GeO₄ is determined.

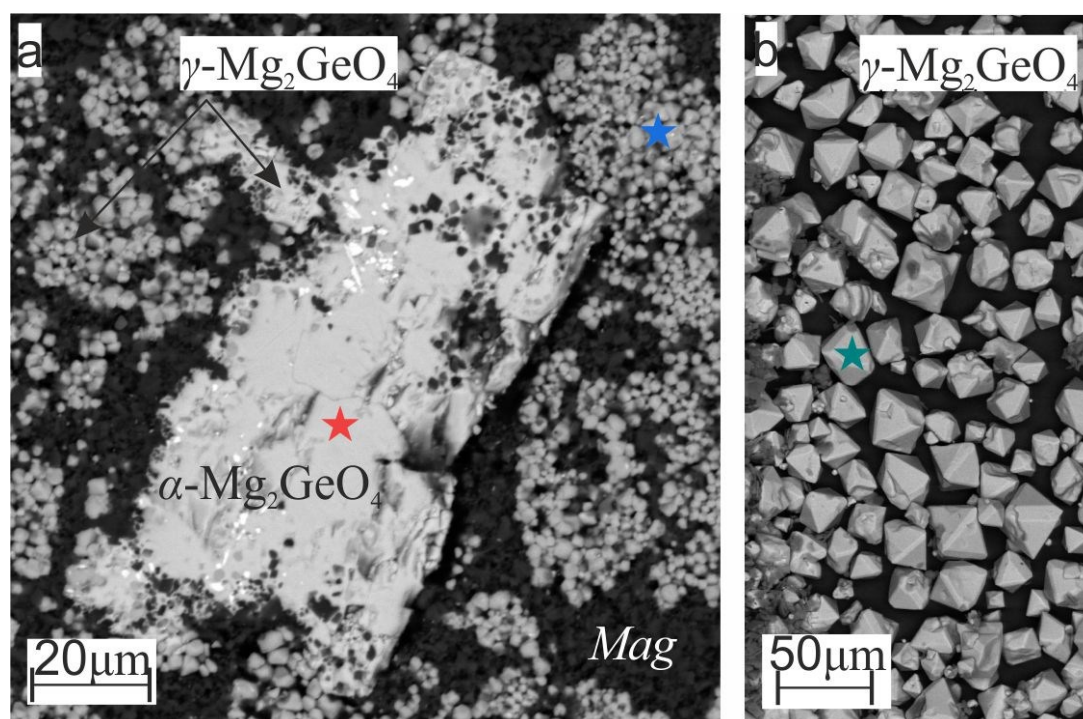


Fig. 1. SEM image of the experimental sample of the (a) $\text{Mg}(\text{OH})_2\text{-MgCO}_3\text{-GeO}_2$ system at 5 GPa and 900 °C and (b) $\text{MgO-GeO}_2\text{-H}_2\text{O}$ system at 7 GPa and 900 °C. Stars in the SEM image mark the points where the Raman spectra were obtained (see fig. 2). *Mag* – magnesite

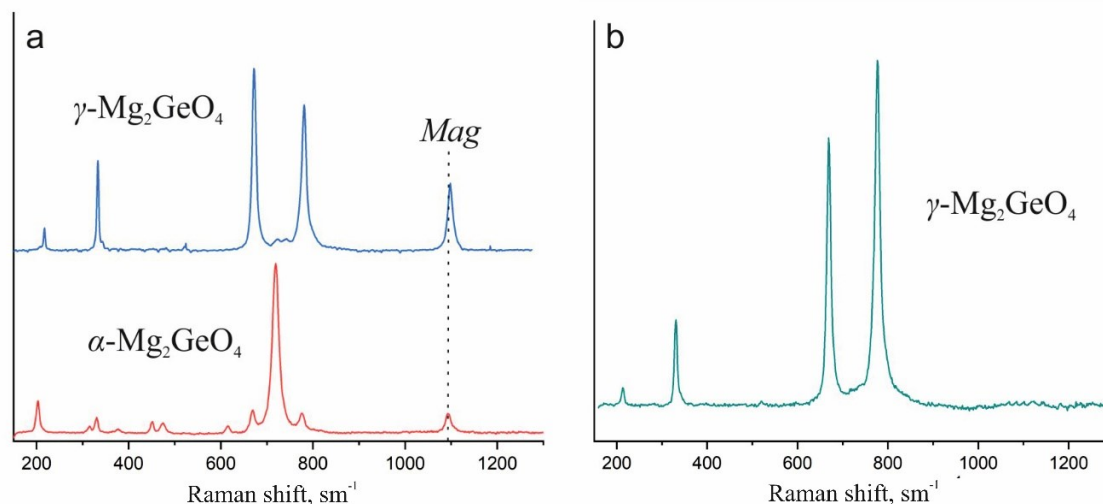


Fig. 2. Raman spectrum of identification of the germanium phase $\alpha\text{-Mg}_2\text{SiO}_4$ and $\gamma\text{-Mg}_2\text{GeO}_4$ from the experimental samples: (a) of the $\text{Mg}(\text{OH})_2\text{-MgCO}_3\text{-GeO}_2$ system at 5 GPa and 900 °C; (b) of the $\text{MgO-GeO}_2\text{-H}_2\text{O}$ system at 7 GPa and 900 °C. The point of obtaining the Raman spectrum is marked with a star in the SEM image (see fig. 1)

Factor group analysis predicts that the olivine structure with $Pnma$ symmetry has 36 Raman active modes: $11A_g + 11B_1g + 7B_2g + 7B_3g$ (Cnopras, 1991). The Raman spectrum of Ge-olivine at standard conditions is characterized by the presence of a low-frequency band group, including lattice vibration modes and deformation modes of the GeO_4 tetrahedron in the range of $150\text{-}520\text{ cm}^{-1}$, and a high-frequency band group at $680\text{-}800\text{ cm}^{-1}$, which correspond to the modes of valence vibrations of the

GeO_4 tetrahedron (Reynard et al., 1994). The topology of the obtained Raman spectrum of $\alpha\text{-Mg}_2\text{SiO}_4$ is similar to those previously published (Guyot et al., 1986; Reynard et al., 1994; Ross et al., 1987), while certain positions of the Raman bands in this work differ slightly from previously published data, which are in a good agreement (Table 1). In addition, the Raman spectrum of Ge-olivine shows a weak inclusion band of magnesite $\sim 1097\text{ cm}^{-1}$ (Fig. 2a red line).

Table 1. Comparison of the positions of the bands of the Raman spectra of α -Mg₂GeO₄, determined in this work, with the literature data, under standard conditions (cm⁻¹).

The positions of the Raman spectrum bands (cm ⁻¹) of α -Mg ₂ GeO ₄			
This work	(Ross et al., 1987)	(Guyot et al., 1986)	(Reynard et al., 1994)
	152		151
	164		163
179			171
203			229
	278		274
315	298		295
330	337		335
342	354		351
378	398		394
	415		412
451	452		450
475	499		496
512	518		515
669	680	681	678
702	688	688	684
719	710	710	706
736	732	731	729
755	762	763	760
775			776
801	802	801	798

Table 2. Comparison of the positions of the Raman bands of γ -Mg₂GeO₄, determined in this work, with the literature data, at standard conditions (cm⁻¹).

The positions of the Raman spectrum bands (cm ⁻¹) of γ -Mg ₂ GeO ₄					Reference
F_{2g}	E_g	F_{2g}	F_{2g}	A_{1g}	
212	330	552	667	776	This work (fig.1)
213	331	520	667	777	This work (fig.2)
213	342	521	671	779	(Thomas et al., 2008)
213	341	520	669	777	(Ross et al., 1987)

The Ge-spinel γ -Mg₂GeO₄ belongs to the spatial group $Fd\bar{3}m$. The factor group analysis suggests the presence of 42 vibrational modes in the spinels: 3 acoustic and 39 optical ones. However, only five Raman-active vibrational modes with the following symmetries $A_{1g} + E_g + 3F_{2g}$ are acceptable for oxide spinels (Setkova et al., 2022). The positions of the Raman bands of synthetic germanium spinel, determined in this work (fig. 2 a blue line, fig. 2 b green line), are shown in Table 2 and are in accordance with previously published works.

As a result of the carried out test experiments in the system Mg(OH)₂ - MgCO₃ - GeO₂ at 5 GPa and 900 °C, α -Mg₂GeO₄ olivine crystals up to 100 microns in size were synthesized. γ -Mg₂GeO₄ spinel crystals up to 50 microns in size were synthesized in the MgO-GeO₂-H₂O system at 7 GPa and 900 °C. The phases were identified using electron probe analysis and Raman spectroscopy. The thermobaric crystallization parameters of the experimentally obtained mineral phases are in accordance with the phase diagrams of the MgO-GeO₂ system. At the

same time, spectroscopic characteristics are a reliable tool for accurate identification of structural modifications (α -Mg₂GeO₄ and γ -Mg₂GeO₄) obtained at various *PT* experimental conditions.

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Verchenko P.A., Setkova T.V., Spivak A.V., Zakharchenko E.S. Synthesis and high-pressure stability of $\text{NaGaGe}_2\text{O}_6$ with pyroxene structure. UDC 549.642

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Abstract. Sodium gallium germanate $\text{NaGaGe}_2\text{O}_6$ was obtained by hydrothermal method at $T = 600^\circ\text{C}$ and $P = 100\text{MPa}$. It adopts the structure of the pyroxene-type chain, with monoclinic symmetry and space group $\text{C2}/c$, unit cell parameters: $a = 9.795(3) \text{ \AA}$, $b = 8.844(2) \text{ \AA}$, $c = 5.4543(16) \text{ \AA}$, $\beta = 104.97(3)^\circ$, $V = 456.5(2) \text{ \AA}^3$. Colourless crystals of elongated morphology, ranging in size from 50 to 300 μm were grown. The spectroscopic characteristics of the new compound in the wavelength range of 100–1200 cm^{-1} have been investigated, including the *in situ* Raman spectroscopy study of the high-pressure behavior of $\text{NaGaGe}_2\text{O}_6$ at pressures up to $\sim 26 \text{ GPa}$. Possible structural transformations were established at pressures of $\sim 6 \text{ GPa}$ and $\sim 19 \text{ GPa}$.

Keywords: pyroxenes; Raman spectroscopy; hydrothermal synthesis; germanium;

Introduction. Pyroxenes are widespread family of rock-forming minerals. The general formula of pyroxenes and their structural analogues can be represented as $M_2M_1T_2\text{O}_6$, where $M_2 = \text{Ca}^{2+}$, Na^+ , and Li^+ ; $M_1 = \text{Al}^{3+}$, Ga^{3+} , Fe^{3+} , Mn^{3+} , V^{3+} , Ti^{3+} , Sc^{3+} , or In^{3+} ; and $T = \text{Si}^{4+}$, Ge^{4+} . The wide isomorphism of cationic sites accounts for chemical and consequently structural diversity among both natural and synthetic compounds. The crystal structure of pyroxenes is characterized by chains of $[\text{SiO}_4]$ tetrahedra oriented along the c -axis and connected by bridging oxygen atoms. Two nonequivalent cationic sites, M_1 and M_2 , are distinguished in the pyroxene structure. Pyroxenes are divided into orthorhombic (orthopyroxenes) and monoclinic (clinopyroxenes) groups, crystallizing in various space groups such as $\text{P2}/n$, Pbca , $\text{P21}/c$, and $\text{C2}/c$.

The synthesis of new compounds with pyroxene structures, including those, that have no natural analogues, as well as the study of their crystal-chemical characteristics and potential properties, is an important task in experimental mineralogy. One of these compounds is the alkali gallium-germanium pyroxene. The compound $\text{NaGaGe}_2\text{O}_6$ was previously synthesized (Redhammer, 2014) at high pressure (20 GPa) and temperature (1200°C). Literature data shows that synthetic analogues of other germanium pyroxenes, such as $\text{NaFeGe}_2\text{O}_6$ (Redhammer et al., 2011), $\text{NaCrGe}_2\text{O}_6$ (Redhammer et al., 2008; Nenert et al., 2009), NaVGe_2O_6 (Emirdag-Eanes & Kolis, 2004), and $\text{NaMnGe}_2\text{O}_6$ (Chen et al., 2013), are also of scientific interest.

The aim of this work is to establish the conditions for hydrothermal synthesis of an alkali

gallium-germanium-containing pyroxene and to conduct a comprehensive study, including under high pressure, which will expand knowledge of the structural variability and potential applications of pyroxene-structured compounds.

Materials and methods. Hydrothermal synthesis of the Ga-Ge pyroxene was carried out at $T = 600^\circ\text{C}$ and $P = 100\text{ MPa}$. The starting mixture consisted of gallium and germanium oxides in a 1:2 molar ratio. Gold capsules with volume of 2 cm^3 were filled with the oxide mixture and an alkali solution (8 wt.% NaOH) according to the filling coefficient (Naumov, 1971), sealed, and placed in a high gas pressure apparatus (HGPA). The duration of the experiments was 8 days.

Crystals obtained by hydrothermal synthesis were analyzed using optical (MBS-10) and polarization (Nikon Eclipse LV100pol) microscopes. Powder X-ray diffraction patterns of the synthesized phase were recorded on a Bruker D8 Advance diffractometer. Unit cell parameters were obtained using a single-crystal X-ray diffractometer Xcalibur AtlasS2, Gemini R-CCD. The chemical composition of the samples was determined on polished surfaces with carbon coating using a Tescan Vega II XMU scanning electron microscope equipped with an INCA Energy 450 energy-dispersive spectrometer (EDS).

Raman spectra at ambient conditions were recorded on a Renishaw RM1000 spectrometer equipped with a Leica microscope and a solid-state

laser at $\lambda = 532\text{ nm}$, at $50\times$ magnification with a 100s acquisition time. Studying the pressure effect on silicates and their germanium analogues is important for understanding Earth's interior. Pyroxenes exhibit various phase transitions depending on composition, temperature, and pressure. Therefore, additional studies of the pressure stability of $\text{NaGaGe}_2\text{O}_6$ were performed by *in situ* Raman spectroscopy. Raman spectra were collected using an Acton SpectraPro-2500i spectrograph with a CCD Pixis2K detector cooled to -70°C and an Olympus microscope with a continuous-wave single-mode solid-state laser at 532 nm. The laser beam was focused on the sample through an Olympus $50\times$ objective to a spot of $\sim 5\text{ }\mu\text{m}$ diameter. The excitation intensity at the sample was $\sim 0.7\text{ mW}$. Signal accumulation time was 540 s ($3\times 180\text{ s}$). A diamond anvil cell with 16-sided anvils and $250\text{ }\mu\text{m}$ culet size was used; the gasket hole diameter was $100\text{ }\mu\text{m}$. NaCl served as the pressure-transmitting medium. Pressure was determined using the ruby fluorescence scale. Raman spectra of the alkali gallium-germanium pyroxene were obtained up to $\sim 26\text{ GPa}$ with 0.25–1 GPa increments. Spectra were processed using Fityk 1.3.1 software.

Results and Discussion. Colourless crystals obtained by hydrothermal synthesis ranged from 50 to $300\text{ }\mu\text{m}$ (Fig. 1a,b). They formed elongated aggregates with prismatic and columnar shapes, exhibiting perfect cleavage along the prism and fracture angles of approximately 90° .

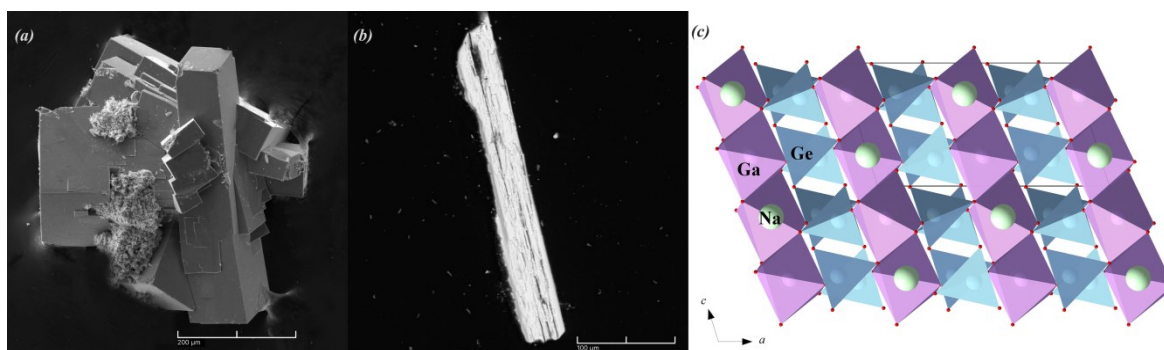


Fig. 1. SEM images of synthesized $\text{NaGaGe}_2\text{O}_6$ crystals as aggregates (a) and single crystal (b); (c) projection of the $\text{NaGaGe}_2\text{O}_6$ crystal structure on the ac plane, visualized in Diamond software based on structural data (Redhammer, 2014).

Phase identification was performed based on powder diffraction data and Raman spectra. According to electron microprobe analysis at 30 points, the chemical composition is uniform throughout the sample with 62.55 wt.% GeO_2 , 28.87 wt.% Ga_2O_3 , and 8.53 wt.% Na_2O . The crystal-chemical formula of the synthesized crystals is close to the ideal $\text{NaGaGe}_2\text{O}_6$. The compound crystallizes in the monoclinic system, space group $C2/c$ (Fig. 1c). Unit cell parameters (Table 1) differ from natural analogues (Cameron, 1973) toward larger values.

This is explained by the incorporation of larger gallium cations ($r_{\text{Ga}} = 0.62\text{ }\text{\AA}$, compared to Al and Fe^{3+} radii of $0.53\text{ }\text{\AA}$ and $0.55\text{ }\text{\AA}$, respectively) in the M1 site and substitution of silicon atoms ($r_{\text{Si}} = 0.26\text{ }\text{\AA}$) by germanium ions ($r_{\text{Ge}} = 0.56\text{ }\text{\AA}$) in the T site. Thus, the unit cell volume of the Ga-Ge analogue is $\sim 13\%$ larger than that of natural jadeite. The parameters slightly differ from those refined by Redhammer (2014), possibly due to different P - T synthesis conditions.

Table 1. Crystallographic characteristics of Ga-Ge pyroxene analogue, aegirine, and jadeite.

NaGaGe ₂ O ₆ (This study)	NaGaGe ₂ O ₆ Redhammer, 2014)	Aegirine NaFeSi ₂ O ₆ (Cameron, 1973)	Jadeite NaAlSi ₂ O ₆ (Cameron, 1973)
$a = 9.795(3) \text{ \AA};$ $b = 8.844(2) \text{ \AA};$ $c = 5.4543(16) \text{ \AA};$ $\beta = 104.97(3)^\circ;$ $V = 456.5(2) \text{ \AA}^3$	$a = 9.9279(9) \text{ \AA};$ $b = 8.8550(8) \text{ \AA};$ $c = 5.4680(6) \text{ \AA};$ $\beta = 107.5419(11)^\circ;$ $V = 458.35(8) \text{ \AA}^3$	$a = 9.658(2) \text{ \AA},$ $b = 8.795(2) \text{ \AA},$ $c = 5.294(1) \text{ \AA},$ $\beta = 107.42(2)^\circ,$ $V = 429.1(1) \text{ \AA}^3$	$a = 9.423(1) \text{ \AA},$ $b = 8.564(1) \text{ \AA},$ $c = 5.223(1) \text{ \AA},$ $\beta = 107.56(1)^\circ,$ $V = 401.8(1) \text{ \AA}^3$

The Raman spectrum of synthetic NaGaGe₂O₆ was obtained for the first time in the 100–1200 cm⁻¹ range (Fig. 2). According to Wang et al. (2001), this range can be divided into five regions: **(I) 1100–800 cm⁻¹** with a strong asymmetric peak near 1000 cm⁻¹ and several broad weak shoulders corresponding to non-bridging tetrahedral stretching modes; **(II) 800–600 cm⁻¹** with an intense doublet or an asymmetric single peak near 670 cm⁻¹, assigned to *T*-O_{br}-*T* stretching/bending modes (where O_{br} is the bridging oxygen connecting tetrahedra in the pyroxene chain); **(III) 600–450 cm⁻¹** with overlapping moderate peaks related to O-*T*-O bending vibrations; **(IV) 450–300 cm⁻¹** with intense overlapping peaks corresponding to internal vibrations of *M*1 and *M*2 octahedra, and tetrahedral chain motion and tilting tetrahedra; **(V) <300 cm⁻¹** with several moderate intensity peaks belonging to lattice vibrations.

The number, positions, and relative intensities of peaks vary for pyroxenes with different elemental compositions. For Ga-Ge pyroxene, main peaks shift to lower wavenumbers compared to natural pyroxenes (Guyot et al., 1986) due to substitution of Si⁴⁺ by heavier Ge⁴⁺ cations. Si–O bond lengths in silicate tetrahedra typically range from 1.60 to 1.65 Å, while Ge–O distances range from 1.70 to 1.80 Å (Wang et al., 2003). Lower vibrational frequencies in germanates compared to silicates are expected due to both increased atomic mass of the tetrahedral cation and weaker Ge–O bond strength (Guyot et al., 1986). Comparative Raman spectra analysis of aegirine NaFeSi₂O₆ (RRUFFID=R040054) and jadeite NaAlSi₂O₆ (RRUFFID=R050220) minerals is presented in Table 2.

Table 2. Wavenumbers (cm⁻¹) of observed bands in aegirine NaFeSi₂O₆, jadeite NaAlSi₂O₆, and NaGaGe₂O₆.

	Jadeite	Aegirine	NaGaGe ₂ O ₆	
V	115.1095 145.5123 204.2876 223.4797	135.0515 162.0819 180.9578 214.8685	113.9578 160.8453 176.0314 187.6897 234.3336	lattice modes
IV	374.8844 386.8974 434.1441	343.0082 388.7049	307.3967 400.5082	O-T-O vibrations internal vibration modes of M1 и M2
III	525.1625 575.677	499.5865 544.0054 554.9259	445.8134 534.2083	O-T-O bending vibrations
II	699.2846 778.9329	678.7038 758.5151	559.5879 772.4261	stretching/bending <i>T</i> -O _{br} - <i>T</i> modes, internal vibration modes of SiO ₄ and GeO ₄
I	885.5667 990.8673 1040.244	868.1086 931.236 973.2798 1044.524	803.2016 831.6752 878.0317	non-bridging tetrahedral stretching modes (SiO ₄ and GeO ₄)

Raman spectroscopy data under pressures up to 26.08 GPa is shown in Fig. 3. The main Raman

bands of NaGaGe₂O₆ shift gradually to higher wavenumbers with increasing pressure. At ~6 GPa,

the band at 159 cm^{-1} disappears, and bands at 113 , 233 , 400 , 559 , and 878 cm^{-1} change their pressure shift slope. At $\sim 19\text{ GPa}$, a new band appears at 425 cm^{-1} , and the band in the 113 – 159 cm^{-1} range disappears. Similar phenomena are observed in some natural pyroxenes such as jadeite (Sakamaki et al., 2012) and spodumene (Jiang et al., 2025). However, no phase transitions up to 8 GPa were detected by *in situ* Raman spectroscopy in natural diopside $\text{CaMgSi}_2\text{O}_6$ and synthetic $\text{CaCoSi}_2\text{O}_6$, $\text{CaCoGe}_2\text{O}_6$, and $\text{CaMgGe}_2\text{O}_6$ pyroxenes (Tribaudino et al., 2016).

Conclusion. Thus, a structural analogue of pyroxene $\text{NaGaGe}_2\text{O}_6$ was grown by hydrothermal

synthesis at relatively low P - T conditions (600°C and 100 MPa). The colorless crystals up to $300\text{ }\mu\text{m}$ in size crystallize in space group $C2/c$ with unit cell parameters $a = 9.795(3)\text{ }\text{\AA}$, $b = 8.844(2)\text{ }\text{\AA}$, $c = 5.4543(16)\text{ }\text{\AA}$, $\beta = 104.97(3)^\circ$, and volume $V = 456.5(2)\text{ }\text{\AA}^3$. The Raman spectra of this compound was obtained for the first time and compared with spectra of natural pyroxenes jadeite and aegirine. Pressure stability of the alkali gallium-germanate was studied in the range 0.11 – 26.08 GPa . Raman spectroscopy indicates possible structural changes in $\text{NaGaGe}_2\text{O}_6$ at pressures of $\sim 6\text{ GPa}$ and $\sim 19\text{ GPa}$.

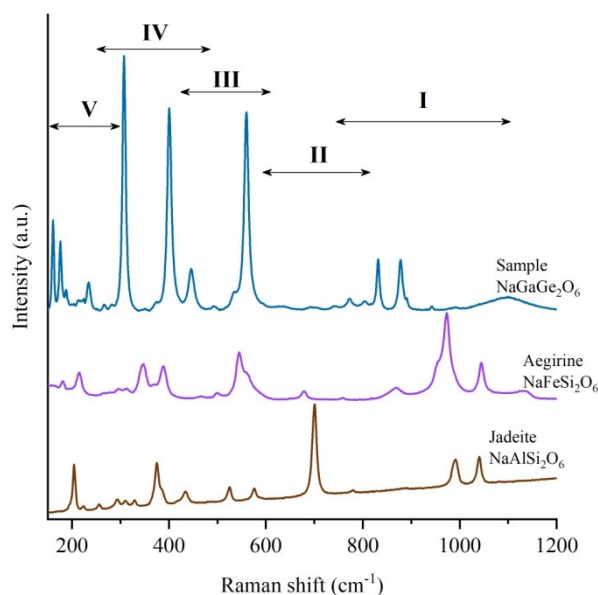


Fig. 2. Unpolarized Raman spectra of synthesized $\text{NaGaGe}_2\text{O}_6$ crystals compared with natural jadeite (R050220) and aegirine (R040054).

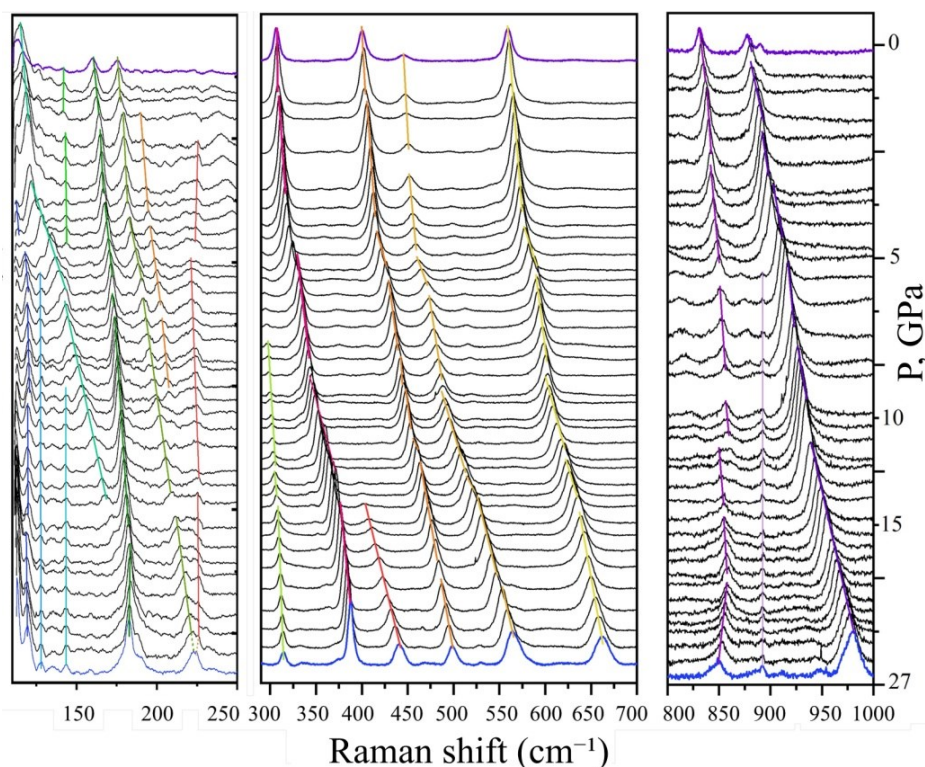


Fig. 3. Unpolarized Raman spectra of synthesized $\text{NaGaGe}_2\text{O}_6$ crystals at pressures from 0.11 to 26.08 GPa .

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